

HIGH-EFFICIENCY LED-BASED ILLUMINATION SYSTEM WITH IMPROVED
COLOR RENDERING

Technical field

The present application is closely related to the following applications:

2003P14657,

2003P14654,

2003P14656.

- 5 The invention is based on a high-efficiency LED-based illumination system with improved color rendering. It deals in particular with luminescence conversion LEDs which in particular are completely tunable.

Discussion of background

One concept for a high-efficiency LED-based illumination system with improved color rendering is three-color mixing. In this concept, the mixture of the primary
10 colors red-green-blue (RGB) is used to generate white. A blue LED can be used for partial conversion of two phosphors which emit red and green. The search for an efficient green phosphor for an RGB system is currently at the forefront of research, as is demonstrated, for example, by the proposal from US 6,255,670. Alternatively, a UV-emitting LED which excites three phosphors respectively emitting in the red,
15 green and blue is used, cf. WO 97/48138. Examples include line emitters, such as YOB:Ce, Tb (green) and YOS:Eu (red). This requires a relatively short-wave emission (UV region < 370 nm) to enable high quantum efficiencies to be achieved. This requires the use of sapphire substrates for the UV-LED, which are very expensive. On the other hand, if a UV-LED based on the less expensive SiC
20 substrates is used, one has to be satisfied with an emission in the range from 380 to 420 nm. The individual colors of the RGB system can in principle be generated by the primary radiation of LEDs or by luminescence conversion LEDs, as is illustrated by WO 01/41215.

To increase the overall light yield, a complicated system including a fourth LED
25 which emits in the range from 575 to 605 nm has also been proposed in

WO 00/19141. By its very nature, a system of this type is significantly more intricate, expensive and complicated than an RGB system.

DE-1 A 101 37 042 shows a slightly different concept, proposing a planar illumination system with special introduction of the blue component in order to avoid the usual absorption problems encountered with blue phosphors.

One interesting class of phosphors for illumination systems of this type are those of the oxynitridosilicate type, as are known per se under the shortened formula $MSiON$; cf. for example "On new rare-earth doped M-Si-Al-O-N materials", J. van Krevel, TU Eindhoven 2000, ISBN 90-386-2711-4, Chapter 6. They are doped with Tb. Emission is achieved under excitation by 365 nm or 254 nm.

A new type of phosphor is known from the as yet unpublished EP patent application 02 021 117.8 (Docket 2002P15736). It consists of Eu- or Eu,Mn-coactivated oxynitridosilicate of formula $MSi_2O_2N_2$ ($M = Ca, Sr, Ba$).

Summary of the invention

It is an object of the present invention to provide an LED-based illumination system with improved color rendering in accordance with the preamble of claim 1 with a color rendering which is as high as possible. A further object is to provide a tunable illumination system.

This object is achieved by the characterizing features of claim 1. Particularly advantageous configurations are to be found in the dependent claims.

The use of RGB-LEDs, which comprise three chips with the emission colors RGB, is of interest for certain applications. Since all three colors are realized by different LEDs, it is possible for all three components to be actuated independently of one another. Therefore, with this type of illumination system it is possible to deliberately set virtually any desired color locus by means of corresponding control electronics. One drawback of this solution is a very low color rendering index $R_a < 50$, which results from the narrow-band nature of the three individual emissions. A further drawback is that the green LED used, for technological reasons, is significantly less efficient than the other two components. An additional factor is that the color locus is

highly dependent on the operating current and the temperature. Current technology (InGaN LED for blue 430 to 470 nm and InGaAlP LED for yellow > 540 nm, in particular red in the range from 600 to 700 nm) has not provided a convincing solution for the primary radiation of an LED to be in the green spectral region.

5 However, the advantage of the RGB solution realized using primary radiation is that illumination systems of this type are tunable.

On the other hand, for illumination systems in which high demands on the color rendering are to the fore, LEDs in which some of the primary LED emission is converted into light with a longer wavelength, in particular green, are used.

10 However, such a structure is not tunable, since the secondary component is not independent of the primary component.

Hitherto, there has not been a high-efficiency green-emitting phosphor which is at the same time insensitive to external influences.

The illumination system according to the invention makes simultaneous use of the
15 color-mixing principle of blue, green and red (RGB mixing) and the principle of converting a primary radiation emitted by an LED into light with a longer wavelength by means of a phosphor which absorbs this radiation, at least two LEDs being used, of which a first LED emits primarily in the range from 340 to 470 nm (peak wavelength), in particular at at least 420 nm, and a second LED emits in the red
20 region at 600 to 700 nm (peak wavelength), wherein the green component is produced by the primary radiation of the first LED being at least partially converted by a green-emitting phosphor, the green-emitting phosphor used being a phosphor from the class of the oxynitridosilicates, having a cation M and the empirical formula $M_{(1-c)}Si_2O_2N_2:D_c$, M comprising Sr as a constituent and D being doped with divalent
25 europium, with $M = Sr$, or $M = Sr_{(1-x-y)}Ba_yCa_x$ with $x+y < 0.5$ being used, the oxynitridosilicate completely or predominantly comprising the high-temperature-stable modification HT.

The invention proposes the use of a phosphor which represents an oxynitridosilicate of formula $MSi_2O_2N_2$ ($M = Ca, Sr, Ba$) which is activated with divalent Eu, if
30 appropriate with the further addition of Mn as co-activator, with the HT phase forming the majority or all of the phosphor, i.e. more than 50% of the phosphor. This HT modification is distinguished by the fact that it can be excited within a broad

band, that it is extremely stable with respect to external influences, i.e. does not reveal any measurable degradation at 150°C, and that it has an extremely good color locus stability under fluctuating conditions (little drift detectable between 20 and 100°C). Further plus points include its low absorption in the red, which is particularly advantageous in the case of phosphor mixtures. This phosphor is often also referred to below as Sr Sion:Eu.

When producing the novel phosphor, it is important in particular to use a high temperature, the synthesis range lying between 1300 and 1600°C. Another determining factor is the reactivity of the starting components, which should be as high as possible.

The phosphor $\text{MSi}_2\text{O}_2\text{N}_2:\text{Eu}$ ($\text{M} = \text{Ca}, \text{Sr}, \text{Ba}$) which is known from EP patent application 02 021 117.8, in the case of the Sr-dominated embodiment with $\text{M} = \text{Sr}$ or $\text{M} = \text{Sr}_{(1-x-y)}\text{Ba}_y\text{Ca}_x$ with $x + y < 0.5$, referred to below as Sr Sion, is difficult to control. Although some test conditions give excellent results, there has hitherto been no guiding principle as to how to obtain desired results in a reliable way. An additional factor is a certain tendency of the efficiency of the phosphor to be reduced and the color locus to vary excessively under high thermal loads.

Surprisingly, it has now been found that the two phases fundamentally differ in terms of their suitability for use as a phosphor. Whereas the LT phase is of only limited use as an Eu-doped phosphor and tends to emit orange-red light, the HT phase has an excellent suitability for use as a phosphor which emits green light. There is often a mixture of the two modifications which manifests both forms of emission over a broad band. It is therefore desirable for the HT phase to be produced in as pure a form as possible, in a proportion of at least 50%, preferably at least 70%, particularly preferably at least 85%.

This requires an annealing process which is carried out at at least 1300°C but no more than 1600°C. A temperature range from approximately 1450 to 1580°C is preferred, since LT phase is formed to an increasing extent at lower temperatures and the phosphor becomes increasingly difficult to process at higher temperatures; above approximately 1600°C it forms a hard-sintered ceramic or melt. The optimum temperature range depends on the precise composition and the properties of the starting materials.

A batch of the starting products which is substantially stoichiometric using the base components SiO_2 , SrCO_3 and Si_3N_4 is particularly important for producing an efficient phosphor of the Sr Sion type. Sr acts as a representative example of M in this context. The deviation should amount to no more than in particular 10%,
5 preferably 5%, from the ideal stoichiometric batch, including any addition of a melting auxiliary, as is often customary. A maximum deviation of 1% is particularly preferred. In addition, there is a precursor for the europium fraction of the doping, realized, for example, as oxide Eu_2O_3 . This discovery runs contrary to the previous procedure of adding the base component SiO_2 in a significantly substoichiometric
10 proportion. This discovery is also particularly surprising on account of the fact that other Sions which are recommended for use as phosphors, such as Ba Sion in accordance with the teaching of EP patent application 02 021 117.8, should indeed be produced with a substoichiometric quantity of SiO_2 .

15 Therefore, a corresponding batch for the Sr Sion $\text{MSi}_2\text{O}_2\text{N}_2$ uses 11 to 13% by weight of SiO_2 , 27 to 29% by weight of Si_3N_4 , remainder SrCO_3 . Ba and Ca fractions in M are correspondingly added as carbonates. Europium is added, in accordance with the desired doping, for example as an oxide or fluoride, as a replacement for SrCO_3 . The batch $\text{MSi}_2\text{O}_2\text{N}_2$ is also to be understood as encompassing any
20 deviations from the exact stoichiometry, provided that they are compensated for with a view to charge retention.

It has proven particularly expedient for the starting components of the host lattice, in particular Si_3N_4 , to have the highest possible purity. Therefore, Si_3N_4 which is synthesized from the liquid phase, for example starting from silicon tetrachloride, is
25 particularly preferred. In particular the contamination with tungsten and cobalt has proven critical. The impurity levels of each of these constituents should be as low as possible, and in particular should in each case be less than 100 ppm, in particular less than 50 ppm, based on these precursor substances. Furthermore, the highest possible reactivity is advantageous; this parameter can be quantified by the reactive
30 surface area (BET), which should be at least $6 \text{ m}^2/\text{g}$, advantageously at least $8 \text{ m}^2/\text{g}$. The level of contamination with aluminum and calcium, based on this precursor substance Si_3N_4 , should as far as possible also be below 100 ppm.

In the event of a deviation from the above procedure with regard to a stoichiometric batch and temperature management, increasing levels of undesirable foreign phases, namely nitridosilicates $M_xSi_yN_z$, such as for example $M_2Si_5N_8$, are formed if the addition of SiO_2 is set at too low a level, so that an excess of nitrogen is produced. Although this compound per se is a useful phosphor, with regard to the synthesis of the Sr Sion, it is extremely disruptive just like other nitridosilicates, since these foreign phases absorb the green radiation of the Sr Sion and may convert it into the known red radiation provided by the nitridosilicates. Conversely, if too much SiO_2 is added, Sr silicates, such as for example Sr_2SiO_4 , are formed, since an excess of oxygen is produced. Both foreign phases absorb the useful green emission or at least lead to lattice defects such as vacancies, which have a considerable adverse effect on the efficiency of the phosphor. The starting point used is the basic principle that the level of the foreign phases should be as far as possible below 15%, preferably even below 5%. In the XRD spectrum of the synthesized phosphor, this corresponds to the requirement that with the XRD diffraction angle 2Θ in the range from 25 to 32°, the intensity of all the foreign phase peaks should be less than 1/3, preferably less than 1/4, particularly preferably less than 1/5, of the intensity of the main peak characterizing the HT modification at approximately 31.8°. This applies in particular to the foreign phases of type $Sr_xSi_yN_z$, in particular $Sr_2Si_5N_8$.

With an optimized procedure, it is reliably possible to achieve a quantum efficiency of from 80 to well over 90%. By contrast, if the procedure is not specific, the efficiency will typically lie in the range from at most 50 to 60% quantum efficiency.

Therefore, according to the invention it is possible to produce a phosphor which represents an oxynitridosilicate of formula $MSi_2O_2N_2$ ($M = Ca, Sr, Ba$) which is activated with divalent Eu, if appropriate with the further addition of Mn as co-activator, with the HT phase forming the majority or all of the phosphor, i.e. more than 50% of the phosphor, preferably more than 85% of the phosphor. This HT modification is distinguished by the fact that it can be excited within a broad band, namely in a wide range from 50 to 480 nm, in particular 150 to 480 nm, particularly preferably from 250 to 470 nm, that it is extremely stable with respect to external influences, i.e. does not reveal any measurable degradation at 150°C in air, and that it has an extremely good color locus stability under fluctuating conditions. Further

plus points include its low absorption in the red, which is particularly advantageous in the case of phosphor mixtures. This phosphor is often also referred to below as Sr Si₃N₈:Eu. A majority of the HT modification can be recognized, inter alia, from the fact that the characterizing peak of the LT modification in the XRD spectrum at approximately 28.2° has an intensity of less than 1:1, preferably less than 1:2, compared to the peak with the highest intensity from the group of three reflections of the HT modification which lie in the XRD spectrum at 25 to 27°. The XRD spectra cited here in each case relate to excitation by the known Cu-K_α line.

With the same activator concentration, this phosphor reveals different emission characteristics than the LT variant of the same stoichiometry. The full width at half maximum of the HT variant is significantly lower in the case of the optimized HT variant than in the case of the simple mixture containing foreign phases and defects, and is in the range from 70 to 80 nm, whereas the simple mixture containing foreign phases and defects has a full width at half maximum of approximately 110 to 120 nm. The dominant wavelength is generally shorter, typically 10 to 20 nm shorter, in the case of the HT modification than in the case of a specimen containing significant levels of foreign phases. An additional factor is that the efficiency of the high-purity HT modification is typically at least 20% higher, and in some cases significantly higher still, than in the case of the LT-dominated mixture or the mixture with a high level of foreign phases.

One characterizing feature of a sufficiently low level of the NT modification and foreign phases is a full width at half maximum (FWHM) of the emission of less than 90 nm, since the lower the level of foreign phases, the lower the proportion of the specific orange-red emission from the modification which is rich in foreign phases, in particular the nitridosilicate foreign phases Sr-Si-N-Eu such as in particular Sr₂Si₅N₈:Eu.

The abovementioned typical reflections in the XRD spectrum, which reveal the different crystal structure, are another important factor, in addition to the reduced full width at half maximum, in establishing the characterization.

The dominant peak in the XRD spectrum of the HT modification is the peak at approximately 31.7°. Other prominent peaks are the three peaks of approximately

the same intensity between 25 and 27° (25.3 and 26.0 and 26.3°), with the peak with the lowest diffraction being the most intensive. A further intensive peak is 12.6°.

This phosphor emits predominantly green light with a dominant wavelength in the range from 550 to 570 nm, in particular 555 to 565 nm.

- 5 It is also possible to add a small amount of the AlO group as a replacement for the SiN group in the molecule of the oxynitridosilicate of formula $\text{MSi}_2\text{O}_2\text{N}_2'$, in particular in an amount of up to at most 30% of the SiN content.

Both phases of the Sr Sion:Eu can crystallize analogously to the two structurally different host lattice modifications and can each be produced using the
10 $\text{SrSi}_2\text{O}_2\text{N}_2$:Eu batch stoichiometry. Minor deviations from this stoichiometry are possible. The Eu-doped host lattices surprisingly both luminesce when excited in the blue or UV region, but in each case after host lattice modification with a different emission color. The LT modification reveals an orange emission, the HT modification a green emission at approximately $\lambda_{\text{dom}} = 560 \text{ nm}$ with in principle a
15 significantly higher efficiency. A desired property of the phosphor can be set accurately as a function of the dopant content and dopant material (Eu or Eu, Mn) and the relative proportions of the HT and LT modifications.

One benefit of the HT phase is the fact that it can be excited with a good level of uniformity over a very wide spectral region with only minor variations in the quantum
20 efficiency.

Moreover, within a wide temperature range the luminescence of the HT modification is only weakly dependent on the temperature. Therefore, the invention has for the first time discovered a green-emitting phosphor, preferably for LED applications, which makes do without special measures to stabilize it. This distinguishes it in
25 particular from the phosphors which have previously been regarded as the most promising candidates for this purpose, namely thiogallate phosphors or chlorosilicates.

The Sion compounds with $M = (\text{Sr}, \text{Ba})$, preferably without Ba or with up to 10% of Ba, represent efficient phosphors with a wide range of emission maxima. These
30 maxima are generally at a shorter wavelength than in the case of pure Sr Sion, preferably between 520 and 565 nm. Moreover, the color space which can be

achieved can be widened by adding small amounts (preferably up to 30 mol%) of Ca and/or zinc; this shifts the emission maxima toward the longer-wave region compared to pure Sr Sion, and by partially (up to 25 mol%) replacing Si with Ge and/or Sn.

- 5 A further embodiment is for M, in particular Sr, to be partially substituted by trivalent or monovalent ions, such as Y^{3+} , La^{3+} or Li^+ , Na^+ . It is preferable for these ions to form at most 20 mol% of the M.

This phosphor has advantages in particular when used in an illumination system, in which case it acts as a green phosphor replacing previous inefficient solutions for
10 the green component. The phosphor is excited either by a blue LED with high-efficiency primary radiation or by a UV-LED. Since the green emission is in a relatively broad band compared to other technological solutions, such as thiogallates or chlorosilicates, a significantly increased color rendering index is established.

This phosphor is particularly well suited to applications in luminescence conversion
15 LEDs which are suitable for full color and luminescence conversion LEDs with colors which can be set as desired based on an LED which primarily emits UV-blue. The conversion by the phosphor according to the invention gives blue-green to green-yellowish colors.

The mixed compounds with $M = (Sr, Ba)$ represent efficient phosphors with a wide
20 range of emission maxima. These maxima are between 520 and 570 nm. Moreover, the color space which can be achieved can be widened by adding small amounts (preferably up to 30 mol%) of Ca and/or zinc and by partially (up to 25 mol%) replacing Si with Ge and/or Sn.

A further embodiment is for M, in particular Sr, to be partially substituted by trivalent
25 or monovalent ions, such as La^{3+} or Li^+ . It is preferable for these ions to form at most 20 mol% of the M.

The phosphor according to the invention can preferably be used for luminescence conversion LEDs to generate white light, albeit with blue primary radiation, but also with UV primary radiation, in which case white light is generated by means of
30 phosphors which emit blue and yellow-green. Candidates for the blue component are known per se; by way of example, $BaMgAl_{10}O_{17}:Eu^{2+}$ (known as BAM) or

$\text{Ba}_5\text{SiO}_4(\text{Cl},\text{Br})_6:\text{Eu}^{2+}$ or $\text{CaLa}_2\text{S}_4:\text{Ce}^{3+}$ or also $(\text{Sr},\text{Ba},\text{Ca})_5(\text{PO}_4)_3\text{Cl}:\text{Eu}^{2+}$ (known as SCAP) are suitable. The phosphor according to the invention is suitable as the yellow-green component.

5 A red phosphor is additionally used to improve the color of this system. It is preferable to employ an additional LED which primarily emits red. It is used in particular together with a blue-emitting base LED, and $(\text{Y},\text{La},\text{Gd},\text{Lu})_2\text{O}_2\text{S}:\text{Eu}^{3+}$, $\text{SrS}:\text{Eu}^{2+}$ or also $(\text{Ca},\text{Sr})_2\text{Si}_5\text{N}_8:\text{Eu}^{2+}$, in particular with a high Ca content, are particularly suitable.

10 It is in this way possible to achieve color rendering index Ra values of from 85 to 95, in particular in wide ranges with warm-white luminous colors corresponding to a color temperature from 2200 to 3500 K, without adversely affecting the dimmability of the illumination system.

15 The solution which has been discovered is therefore now superior to both sub-aspect solutions, since it allows a higher efficiency than the previous efficiency-optimized system to be achieved and also offers a much better solution to the dimmable system. The overall result is a breakthrough in this technology.

20 Particular preference is given to an illumination system in RGB technology which uses only nitride-based phosphors by using a high-efficiency blue LED with a dominant wavelength from 440 to 465 nm, preferably with a peak wavelength of 460 nm, together with luminescence conversion LEDs. A first luminescence conversion LED uses a blue LED, preferably with a peak wavelength of 460 nm, as primary light source, with conversion by means of the above-described Sr Sion as green secondary light source. A second luminescence conversion LED uses a blue LED, preferably with a peak wavelength of approximately 460 nm, as primary light
25 source, with conversion by means of a nitridosilicate of type $(\text{Ca},\text{Sr})_2\text{Si}_5\text{N}_8:\text{Eu}^{2+}$ as red secondary light source. Surprisingly, these three components virtually ideally complement one another in their spectrum, thereby allowing a high color rendering at high efficiency.

30 The technical realization of the illumination system according to the invention can be implemented in various ways. In particular, what are known as multichip LEDs are of

interest, in which the various chips are located inside a housing. There are generally two or three chips. In principle, the following implementations are possible:

In a first embodiment, the first LED is a UV-LED with primary emission in the range from 340 to 430 nm, which excites the green phosphor to secondary emission. The
5 second LED is the red-emitting LED. In addition, a third LED is used, which preferably itself emits primarily in the blue (430 to 470 nm peak) or also in which a blue phosphor is excited by an LED which emits primarily UV.

In a second embodiment, just two LEDs are used. In this case, the first LED either emits primarily UV in the range from 340 to 420 nm, in which case a blue-emitting
10 phosphor and the novel green-emitting phosphor are arranged ahead of it. These two phosphors convert the UV radiation of the first LED completely. However, it is preferable to use an embodiment in which the first LED is a blue-emitting LED with a peak in the range from 430 to 470 nm, which is assigned a novel green-emitting phosphor which partially converts the primary light of the LED into green secondary
15 radiation. The second LED is once again the red-emitting LED. Of course, the possibility of the red component also being generated by conversion of a radiation with a shorter wavelength, for example from a UV-LED or blue LED, is not ruled out.

Naturally, the LEDs described here may also be understood as encompassing groups of LEDs of the same type.

20 The individual chips are in this case locally provided with the respective phosphor. For this purpose, the individual chips may be located in different hollows or cavities or together in a single cavity. The chips have in this case normally already been provided with the phosphor in a prior process. In the case of the solution with a single cavity or a solution in which the phosphor is arranged so as to be physically
25 separate from the chip, it is also possible for the phosphor to be applied only after the chips have been installed in the housing of the illumination system. The phosphors described are particularly suitable for technologies using near-chip conversion, as are known per se from the literature, cf. for example DE 102 03 795.

Furthermore, the invention relates to an illumination system having LEDs as
30 described above, the illumination system also including electronic components which, by way of example, impart dimmability. A further purpose of the electronics is

to actuate individual LEDs or groups of LEDs. These functions may be realized by known electronic components.

Figures

The invention is to be explained in more detail in the text which follows on the basis of two exemplary embodiments. In the drawing:

- 5 Figure 1 shows an emission spectrum for an oxynitridosilicate;
 Figure 2 shows the reflection spectrum of this oxynitridosilicate;
 Figure 3 shows an illumination system having a plurality of semiconductor
 components which serves as light source for white light (fig. 3b), with
 a semiconductor component also being shown on a larger scale
10 (fig. 3a);
 Figure 4 shows an emission spectrum for the semiconductor component from
 figure 3;
 Figure 5 shows a second exemplary embodiment of a semiconductor
 component.

15

Description of the drawings

Figure 1 shows a specific example for the high-efficiency, green-emitting phosphor. This example relates to the emission of the phosphor $\text{SrSi}_2\text{N}_2\text{O}_2:(10\% \text{Eu}^{2+})$ in the HT modification, in which the Eu fraction forms 10 mol% of the lattice sites occupied by Sr. The emission maximum is at 545 nm, the mean dominant wavelength at
20 564 nm (λ_{dom}). The color locus is $x=0.393$; $y=0.577$. The excitation takes place at 460 nm, and the FWHM is 84 nm.

Figure 2 shows the diffuse reflection spectrum for this phosphor. It reveals a pronounced minimum in the range below 430 nm, which therefore demonstrates the good excitability in this range.

25 Figures 3a, 3b specifically illustrate the structure of a light source for white light. The light source, cf. figure 3b, is a semiconductor component 6 of the LED type having a first chip 1 of the InGaN type with a peak emission wavelength of, for example,

460 nm, and a second chip 2 of the InGaAlP type with a peak emission wavelength of, for example, 620 nm, and finally a semiconductor component of the luminescence conversion LED type with a third chip 3 of the InGaN type with a primary peak emission wavelength of, for example, 460 nm. The semiconductor component 6, together with other similar elements, is embedded in an opaque basic housing 8. The phosphor is the oxynitridosilicate $\text{SrSi}_2\text{O}_2\text{N}_2:\text{Eu}(10\%)$ proposed as the exemplary embodiment, which completely converts the primary radiation of the chip 3, transforming it into green radiation with a peak emission at 547 nm with $\lambda_{\text{dom}} = 563$ nm. This solution has the major advantage of being tunable within a wide range of color temperatures by changing the relative intensities of the three LEDs by means of electronic control unit 7. A comparison, cf. Table 1, with the solution which has hitherto been available using three primary emitting LEDs (RGB, with green realized by an InGaN LED with $\lambda_{\text{dom}} = 526$) convincingly demonstrates the superiority of the new solution. Fig. 3a shows a view of an LED 6 on a larger scale.

Figure 4 shows the emission from an illumination system of this type as spectral distribution (intensity in arbitrary units) against the wavelength (in nm). The dashed line shows the old solution (three primary emitting LEDs) compared to the new solution (two primary emitting LEDs and a luminescence conversion LED for green) for a color temperature of 4000 K.

The particular advantage of using a long-wave primary light source (450 to 465 nm) for the green luminescence conversion LED is that this avoids problems with ageing and degradation of the housing and resin or phosphor, with the result that a long service life is achieved.

In another exemplary embodiment, a UV-LED (approximately 380 nm) is used as primary light source for the green luminescence conversion LED; in this case, problems with ageing and degradation of housing and resin or phosphor have to be avoided as far as possible by means of additional measures which are known per se, such as careful selection of the housing material, addition of UV-resistant resin components. The major advantage of this solution is the very high efficiency of typically 30 lm/W which can thereby be achieved.

Table 1: Comparison of the color rendering index Ra and of the red index R9 between white-emitting semiconductor components based on the pure LED solution (old) and the solution using a green luminescence conversion LED (new)

Color temperature (K)	Ra (old)	R9 (old)	Ra (new)	R9 (new)
2700	38	-23	91	92
3000	38	-37	91	93
4000	43	-71	94	89
5000	36	-87	91	78
6430	51	-99	86	57

- 5 In a further exemplary embodiment, cf. figure 5, a solution with two LEDs is used as white-emitting semiconductor component. The basic structure is similar to that described in WO 01/41215. A first luminescence conversion LED provides the blue and green components. A chip 1 of type InGaN with a primary peak emission wavelength of, for example, 460 nm is embedded in an opaque basic housing 8 in
10 the region of a cavity 9. At the same time, a second LED 2 of type InGaAlP, which emits red light, similar to the first embodiment, is also accommodated in the cavity 9.

- The chips have separately controllable, separate terminals 3. In each case one of the terminals 3 is connected to the chip 1, 2 via a bonding wire 4. The recess has an inclined wall 7 which serves as reflector for the primary radiation of the chips 1, 2.
- 15 The recess 9 is filled with a potting compound 5, which as its main constituents typically contains a silicone casting resin (or alternatively epoxy casting resin) (80 to 90% by weight) and phosphor pigments 6 (less than 15% by weight). Further small fractions can be attributed, inter alia, to methyl ether and Aerosil. The phosphor is the oxynitridosilicate $\text{SrSi}_2\text{O}_2\text{N}_2:\text{Eu}(10\%)$ proposed as the first exemplary
20 embodiment in a lower concentration, which only partially converts the primary radiation of the LED, transforming it into green radiation with a peak emission at 540 nm, or $\lambda_{\text{dom}} = 560 \text{ nm}$.

This narrow construction with a common cavity is possible since the red LED 2 with primary emission at 645 nm is not absorbed or converted by the green phosphor. This shows an example of the significance of a narrow full width at half maximum (FWHM below 90 nm, preferably below 80 nm). The only drawback of this extremely compact solution, which has been proposed for the first time here, compared to the three LED solution is the lack of tunability.

The illumination system is in particular also suitable for the concept of adaptive illumination, in which the luminous color or also the brightness of the illumination system can be set according to predetermined criteria which can be selected as desired or is suitably matched to the brightness of the surroundings.